Dielectric Studies of Lead Titanate Prepared by Wet Chemical Method

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Abstract:

Solution of molar proportion of precursors Lead Nitrate $[Pb(NO_3)_2]$, Titanium Isopropoxide $[C_{12}H_{28}O_4Ti]$ in liquid, and pure water stirred for 1 hr at 100°C by using the wet chemical route. The white raw powder so obtained is then calcined at different temperature at 500°C, 550°C and 600°C for 4 hours. Thus obtained nanoparticles of Lead Titanate PT (PbTiO_3) were subjected to X-ray diffraction [XRD] studies. XRD of the lead titanate reveals 32.39 nm, 32.04 nm and 30.87 at 500°C, 550°C and 600°C calcinations. **Key words**: Wet Chemical Route, PbTiO_3, XRD, Dielectric properties.

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I. Introduction:

Perovskite systems (ABO₃) have been studied widely for its B site substitutions. Basically, PbTiO₃ has been considered to be one of the most important members of this family [1]. It has a high Curie temperature, high pyroelectric coefficient, low dielectric constant, and high spontaneous polarization. Lead titanate (PbTiO₃, PT) is a ferroelectric nanocrystalline powder that has not only been proved to be a technologically important material but also it is a significant component material in electronics such as capacitors, ultrasonic transducers, thermistors, and optoelectronics [2]. These systems are due to their potential applications as ferroelectric materials. The improvement in the piezoelectric and pyroelectric properties is also described by replacing Pb in the A site of the ABO₃ structure [3-4]. Large c/a ratio in PbTiO₃ at low temperature confer tetragonal phase, it disintegrates into powder when cooled through the Curie point. Synthesis of PbTiO₃ by wet chemical methods offer advantages because of high-purity, good stoichiometry and controlled particle size [5-9].

The conventional solid-state reaction has a tendency to produce a coarse PbTiO₃ powder with compositional in homogeneity and a degree of particle agglomeration if the processing parameters are not carefully optimized [10-14]. Therefore, many chemistry-based processing routes, including co-precipitation, solgel synthesis, hydrothermal and citrate routes have been devised for the preparation of an ultrafine, sintering-reactive PbTiO₃ powder. Our method of synthesizing PbTiO₃ relies on the reaction between Lead Nitrate [Pb(NO₃)₂] and Titanium Isopropoxide [C₁₂H₂₈O₄Ti] at high temperature [15-18].

Experimentation:

Molar proportions of Pb(NO₃)₂ (Lead Nitrate) is dissolved in Titanium Isopropoxide $[C_{12}H_{28}O_4Ti]$ in liquid. The lead titanate were prepared by dissolving solid lead nitrate powder in pure water and stirred for 1 hr at 100° C in reaction flask, when the lead nitrate was dissolved in water, then a stoichiometric amount of titanium Isopropoxide was added to the solution and the solution was refluxed at 100 °C for 1 hr. and the solution was kept at 100 °C for 12 hr. to get nanocrystalline powder. Nanocrystalline powders with various crystallite sizes were obtained by calcining the powder at different temperatures at 500°C, 550°C and 600°C for 4 hr. Structure and phase transformation analyses were investigated by XRD in the range of 5°– 90°. The average crystallite size was calculated from the full width at half maximum of the diffraction lines using Scherrer's relation.

X-RAY DIFFRACTION:

II. Results And Discussions

XRD pattern of the PbTiO₃ calcined at 500°C, 550°C and 600°C is shown in Fig.1 at 500°C; the crystallization of tetragonal lead titanate phase began along with traces impurity phase. At 550 °C all peaks can be indexed to the PbTiO₃ with a tetragonal structure. No peak corresponding to any of the source materials or allotropic forms was found, suggesting that a pure crystalline compound exists. The crystallite size of the particles calcined at various temperatures could be calculated by the Scherrer's equation: $t = k\lambda/\beta cos\theta$ (where t

is the average size of the particles, assuming particles are spherical, k = 0.9, λ is the wavelength of X-ray radiation, β is the full width at half maximum of the diffracted peak and θ is the angle of diffraction. The crystallite size obtained from XRD data at 500°C, 550°C and 600 °C were 32.39 nm, 32.04 nm and 30.87 nm respectively. The peak related to the plane 100 is getting suppressed with the elevated sintering temperature. This peak belongs to the excess amount of PbO present in the material. As sintering temperature is increased, the peak belonging to PbO is vanished and pure phase PbTiO₃ is obtained.





Fig.5 Sintering temperature vs. Density

Fig.2. shows the variation of c/a ratio with the sintering temperature, which clearly indicates c/a ratio, is maximum at 500°C which means stable tetragonal phase at 500°C and decreases for 550°C and 600°C. Decrease in c/a ratio afterwards indicates dispersion from tetragonality and an approach to the cubic phase with higher sintering temperature. Fig.3 shows the variation of average grain size with sintering temperature indicating decrease in average grain size with increasing sintering temperature. The density of the material is initially increases and then decreases slightly owing to the structural changes during calcinations. The change in the density is so small that it can be inferred that the density of the material is almost constant in the given calcinations temperature range.

Dielectric Properties:

The dielectrics properties of PT nanoparticles were studied to determine various electrical parameters with respect to frequency. Figure 6.shows the dielectric constant (k) of the PT nanocrystalline powders as a function of frequency at room temperature. The dielectric constant decreases with respect to frequency. The decrease in frequency is rapid in low frequency region and becoming constant in high frequency region representing dielectric dispersion. The dielectric dispersion with frequency shown by the material is due to Maxwell-Wagner type interfacial polarization. The large value of dielectric constant at low frequency is attributed to the presence of all types of polarization, whereas at higher frequencies the dominant contribution to dielectric constant is from electronic polarization only. Fig.6 shows variation of the dielectric constant with respect to calcinations temperature and this may be due to evaporation of oxide ions at higher calcinations temperature.



Fig.3 Sintering temperature vs. Dielectric constant

III. Conclusion:

This work done attempted to generalize the reaction for material synthesis by using the wet chemical rout is successful. The average crystallite size 32.39 nm, 32.04 nm and 30.87 at 500°C, 550°C and 600°C calcinations was assured from XRD peaks. The variation in c/a ratio showed transition from tetragonal to Cubic phase. The grain size also decreases with increased calcinations temperature. Cell volume of the material increases with increased calcinations temperature. Cell volume of dielectric constant at low frequency attributed to the presence of all types of polarization, whereas at higher frequencies the dominant contribution to dielectric constant is from electronic polarization only. With increased sintering temperature, charge holding capacity of the material decreases owing to the evaporation of ions due to oxidation.

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