

Effect of Doping Concentration on the Structural Properties of Zn: SnO₂

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Abstract: Undoped tin Oxide and Zinc doped tin oxide thin film were prepared by Chemical spray pyrolysis on soda lime glass substrate. The effects of doping concentration (1wt%, 2wt%, 3wt% and 4wt%) on the structural properties of the films were investigated. X-ray diffraction analysis indicates that the films were polycrystalline, having tetragonal rutile structure irrespective of their doping concentration and exhibits a preferred orientation along the (101) plane while the crystallinity was enhanced with increased in doping concentration to an optimum value at 4 wt %. The structural parameters such as particle size, strain and dislocation density were affected and calculated. The particle size of SnO₂ is 27.9 nm while that of Zn:SnO₂ thin film was found to be within the range of 26.2 nm to 11.2 nm. The thickness of the film decreases with increase in doping concentrations within the range of 0.030 to 0.018 μm.

Key words: TCO, Thin film, Doping.

I. Introduction

Transparent conducting oxides (TCOs) are solid-state oxides with low resistance and high transparency in the visible range of the electromagnetic spectrum. Most inorganic films typically are made up of a layer of transparent conducting oxides, generally in the form of Indium doped Tin IV Oxide (ITO), Fluorine doped Tin IV Oxide (FTO), Zinc doped Tin IV Oxide (ZTO), [1]. TCOs are wide band-gap semiconductors that have relatively high concentration of free electrons in their conduction band. These arise either from defects in the material or from extrinsic dopants, the impurity levels of which lie near the conduction band edge. Reduction of the resistivity involves either an increase in the carrier concentration or in the mobility [2]. The high-electron-carrier concentration causes absorption of electromagnetic radiation in both the visible and infrared portions of the spectrum. For the present purposes, it is the former that is the more important.

Transparent conducting oxides have band gaps with energies corresponding to wavelength which are shorter than the visible range of 380 nm to 700 nm. As such, photon with energies below the band gaps are not collected by these materials and these visible lights passes through [3].

Tin dioxide (SnO₂) is an n-type broad-band gap (3.6eV) oxide semiconductor with high chemical and mechanical stability. SnO₂ which has exceptional optical, electrical and mechanical properties because of the low resistivity and high transmittance, It is a multipurpose material and has wide usage as the most smart material for gas sensor applications, as a catalyst during the oxidation of organic compounds, as a crucial component in rechargeable Li batteries and as a master element in opto-electronic devices [4]. Tin (IV) Oxide is an inorganic compound. The mineral form of SnO₂ is called cassiterite and this is the main ore of tin. Tin dioxide is white, diamagnetic and amphoteric in nature. It is also a non-flammable substance.

However such Zinc dopant requires a high degree of deposition control or the optimization of dopant concentration to obtain a carrier concentration suitable for use as a TCO in thin film solar cell due to the high reactivity of SnO₂ with oxygen. Above a certain level, the excess Zn atoms exist in the films as interstitial atoms or tend to segregate into grain boundaries and poor crystallized area there by working as electrically inactive sites with a build-up of an energy barrier limiting carrier transport and decreasing carrier concentration. As such we investigate the effects of doping concentration on the structural properties of Tin dioxide (SnO₂) doped with Zinc thin film.

II. Materials And Methods

2.1 Deposition Parameters

In this work, the quality of films deposited depends on the following deposition parameters; the substrate temperature, spray rate and the height of the nozzle from the substrates. Chemical spray pyrolysis technique depositor of K-type thermocouple is used. Rate of flow is 1.92 ml/min and height of nozzle from substrate is 12cm. When the substrate temperature is below 290 °C, the spray falling on the substrate will produce foggy film as a result of incomplete thermal decomposition (oxidation) as a result; will give rise to a film whose transparency as well as electrical conductivity is very poor. When the substrate temperature is in the range of 400 – 500 °C the spray reaches the substrate surface in the semi vapour state and complete oxidation

takes place forming clear SnO₂ thin film. If the substrate temperature is greater than 500⁰C the spray gets vaporized before reaching the substrate and the film becomes almost powdery. As such in this work a temperature of 420⁰C was used for the deposition.

2.2 Fabrication of Undoped SnO₂

The fabrication of the undoped Tin dioxide (SnO₂) was carried out using Tin (IV) Chloride (SnCl₄) solution as the source element. During the fabrication of undoped tin (IV) Oxide, 10ml of Tin (IV) Chloride solution, and 10ml of distilled water were added to the solution container of the spray pyrolysis equipment after which it was switched on for the thin film deposition to begin. The thin films of Tin (IV) Oxide (SnO₂) were deposited on a glass substrate. During the fabrication of the doped thin film, 10ml of the doped Tin (IV) Chloride solution with zinc acetate (Zn(CH₃COOH)₂·2H₂O) as the dopant and 10 ml of distilled water was used and the thin film form of tin (IV) Oxide doped with zinc, (Zn.SnO₂) were deposited on a glass substrate. The temperature of thermocouple of the chemical spray pyrolysis depositor set-up which was kept constant throughout the fabrications process was 420⁰C and the room temperature was 30⁰C. The distance between the substrate and the spray nozzle was fixed at 12 cm through the fabrication process. The consumption rate or the spray rate was 1.92 ml/min. The time of deposition of the film on the glass slide was five minutes because at that time, quality thin film, good for characterisation have already been deposited on the glass slide.

2.3 Fabrication of Tin (IV) Oxide Doped With Zinc

Tin (IV) Oxide (SnO₂) doped with Zinc was fabricated by doping from 1wt.% to 4 wt.% of Zn in the source material (SnO₂) using the relation;

$\frac{\text{Zn (g)}}{\{\text{Zn (g)} + \text{Sn (g)}\}}$ = 1%, 2%, 3% and 4% of the Zn by weight used for the doping. 1000ml SnCl₄ solution used contained 36.83g of SnCl₄; this implies 10ml of SnCl₄ solution used for the fabrication contain $\frac{10}{1000} \times 36.83 = 0.368$ g.

Table 1 SnO₂ doped with Zinc

Wt. of Sn (g)	Wt. of Zn(g)	% of Zn	Zinc Acetate(ml)	SnCl ₄ (ml)
0.37	0.0037	1	0.269	9.731
0.37	0.0074	2	0.533	9.467
0.37	0.0111	3	0.790	9.210
0.37	0.0148	4	1.045	8.955

III. Results And Discussion On Surface Thickness Of The Film

The surface thickness of the films was carried out using Profilometer with stylus of 12.5 μm, length of 2000 μm, resolution of 0.333μm and duration of 10.0 second. The results are shown in table 2 for the undoped SnO₂ and 1 wt.% to 4 wt.% Zn doped SnO₂ respectively. The values of the thickness of the thin films are then used to calculate the resistivity values of the thin film. Table 2 gives the result of the thickness of the thin film.

Table 2 Shows the Result of thickness for the thin film

Sample	Thickness of thin film (μm)
Undoped SnO ₂	0.150
1 wt.% Zn doped SnO ₂	0.030
2 wt.% Zn doped SnO ₂	0.025
3 wt.% Zn doped SnO ₂	0.020
4 wt.% Zn doped SnO ₂	0.018

Table 2 shows the result of the thickness of the undoped SnO₂ thin film and the 1 wt.% to 4 wt.% Zn doped SnO₂ thin film. It was observed that the thickness of the thin film decreases with increase in doping concentration of Zn. This is due to the decrease in grain size of the thin film which as a result of increase in the grain boundary of thin film. The decrease in the thickness of the thin film is also attributed to Beer Lambert's law which shows that as the transmittance increase the thin film thickness decrease at wavelength range of the visible portion of the electromagnetic spectrum between 400 nm and 780 nm for transparent conducting oxides.

3.1 Discussion on XRD analysis

The X-ray diffraction pattern of the thin film is recorded in Figure 1. From the result, it shows that both the doped and the undoped Tin (IV) Oxide (SnO₂) thin film have tetragonal rutile-type structure with lattice parameters $a = 4.7370\text{Å}$, $c = 3.1850\text{Å}$. The XRD pattern was analyzed with Gaussian function where full width half maxima [FWHM] were determined. Undoped film were found to exhibit six diffraction peaks associated in figure 1 with (110), (101), (200), (211), (220) and (310) planes. The 1% Zn doped SnO₂ were found to exhibit five diffraction peaks with (110), (101), (111), (211) and (301) planes. The 2% Zn doped SnO₂ were found to exhibit six diffraction peaks with (110), (101), (111), (211), (221) and (301) planes. The 3% Zn doped SnO₂

were found to exhibit two diffraction peaks with (101) and (200) planes. The 4% Zn doped SnO₂ were found to exhibit six diffraction peaks with (110), (101), (111), (211), (310) and (221) planes.

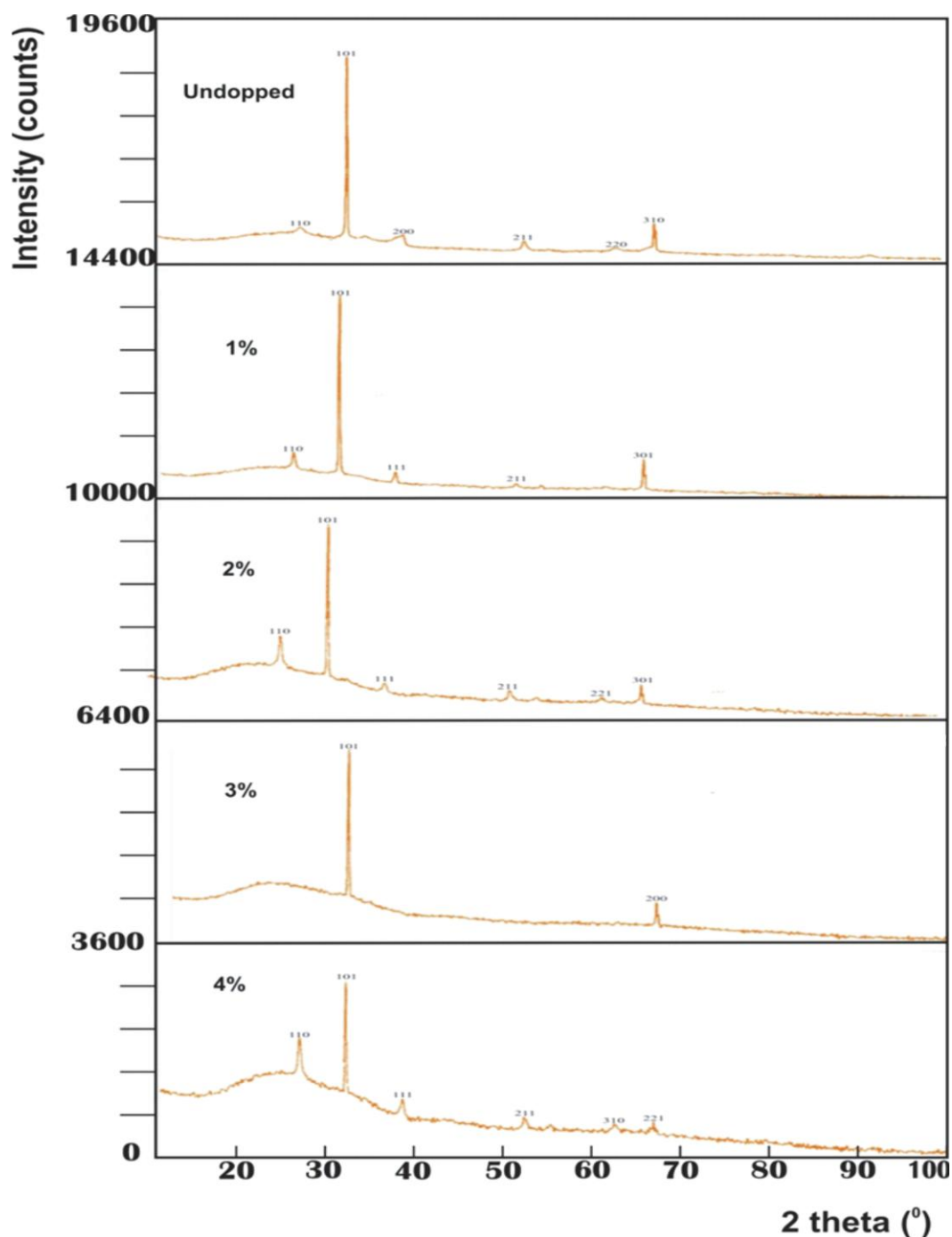


Figure 1: XRD patterns of undoped SnO₂ and Zn doped SnO₂ thin film

From figure 1 it was observed that deposited films have preferred orientation along (101) plane in all the Zn doped SnO₂ films deposited at doping concentration 1wt.% to 4 wt.%, this is because the 101 plane has relatively high intensity for all the samples irrespective of the doping concentration. The low intensity and a broad diffraction peak were identified, corresponding to an early crystallization of the Zn doped SnO₂ in the oriented structure and such peaks are, (111), (211), (221), (200) and (310) as shown in figure 1 this decrease in the peak intensity is also attributed to the increase in the number of defects or reduction in the atomic weight as a result of the substitution of the Sn²⁺ ion (0.071Å) with the Zn²⁺ ion (0.074Å). As the doping concentration increased, the increase in peak intensity is associated with 101 plane, the narrowness of the (101) diffraction peak indicates growth of the (101) oriented crystallites. At 4 wt.%, there are very small diffraction at around 70°

and 80° 2θ scan whose match were not found. The decrease in peak intensity which results to broadness in diffraction peaks might also be due to change in electron density in the conduction band or due to point defect. The crystal structure of SnO₂ did not change because the lattice parameters were not change by the Zn doping, the reason is, Sn⁺⁴ ion have close radii to the Zn⁺² ion, that is; the ionic radius of Sn⁺⁴ is r = 0.71 Å and Zn⁺² is r = 0.74 Å. This might also be due to the substitution of metal ions in the crystallite structure and therefore result to no structural strain in the Zn:SnO₂

3.2 Discussion on grain size

The grain size of the thin film was calculated using Debye Scherer formula

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (1)$$

Where D is the grain size, λ is wavelength of the CuKα radiation source (1.54060 Å) wavelength, β is the Full Width Half Maxima (FWHM) and θ is the diffraction angle.

Table 1: Shows the Result of grain size for the thin film.

Sample	Grain size of thin film (nm)
Undoped SnO ₂	27.9
1 wt.% Zn doped SnO ₂	26.2
2 wt.% Zn doped SnO ₂	22.3
3 wt.% Zn doped SnO ₂	17.9
4 wt.% Zn doped SnO ₂	11.2

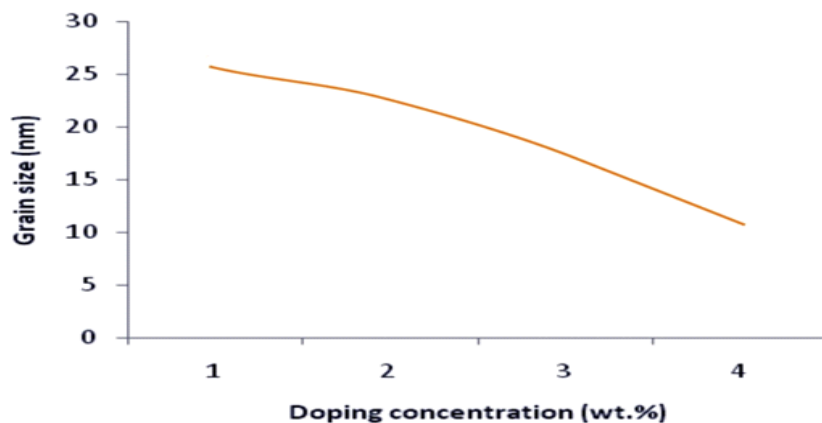


Figure 2: graph of grain size against doping concentration.

From figure 2 Increasing in doping concentration to 4 wt.% causes decrease in the grain size, this is because increase in doping concentration increases the FWHM of the diffraction peak which is an indication of decrease in particle size. This is a clear reason for the decrease in peak intensity with increase in doping concentration. The grain size could also have decrease with increase in doping concentration due to the effect Zn ion on the nucleation during the process of film growth.

3.3. Discussion on SEM analysis

The Scanning Electron Microscopy (SEM) characterisation reveals that all the SnO₂ thin films are almost uniform. The SEM micrographs of pure SnO₂ thin films and 1 to 4 wt.% Zn doped SnO₂ is shown in figure 3 and figure 4

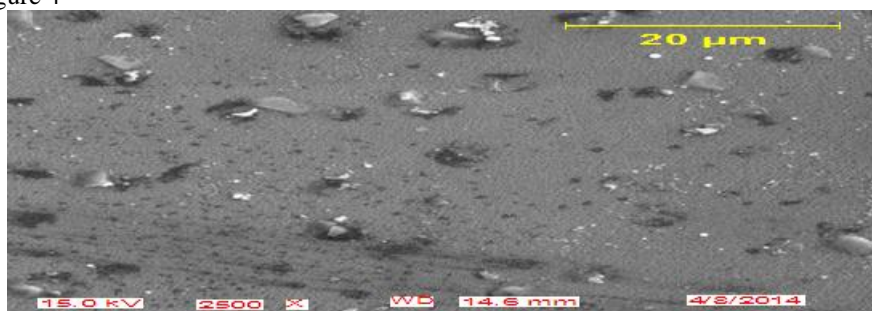


Figure 3: Scanning Electron Micrograph (SEM) of SnO₂

The SEM pictures in figure 3 exhibit the view of the thin film surface which consists of very thin network of SnO₂ nanoparticles. Surface morphology of the SEM micrograph showed fine tiny nanoparticles, with certain clustering.

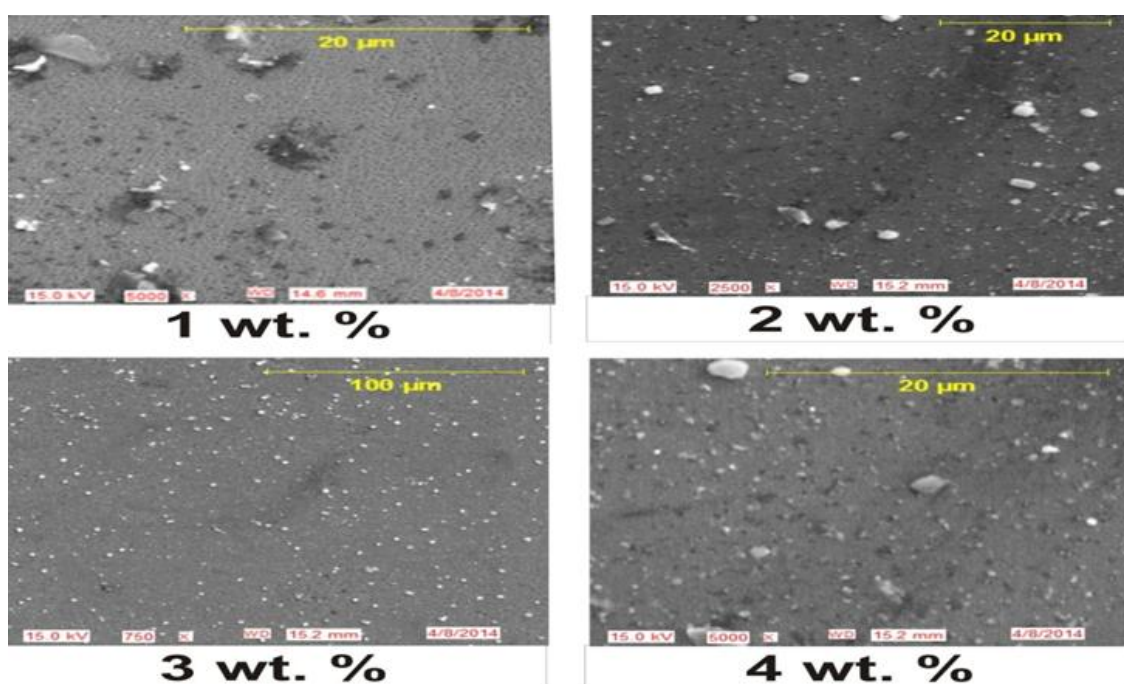


Figure 4: Scanning Electron Micrograph of Zn doped SnO₂

The SEM pictures in figure 4 consist of very thin network of Zn doped SnO₂ nanoparticles. With increase in doping concentration, the SEM micrographs of the thin films were found to be almost uniform. This uniformity in the SEM micrograph is due to the fact that on substitution of Zn ion in the crystal structure of the SnO₂ thin film, no structural strain was generated because of the closeness of the ionic radius of Zn and Sn atom.

IV. Conclusion

On doping Tin IV Oxide with Zinc at doping concentration of 1 wt.% to 4 wt.%, the XRD result shows the plane 101 to have the most preferred orientation at all doping concentration. The thickness of the doped Tin IV Oxide decrease with increase in doping concentration and the SEM micrographs of the thin films were found to be almost uniform at all doping concentration.

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